

ELECTRONIC DEVICE WITH ELECTRODE AND ITS MANUFACTURE

CROSS REFERENCE TO RELATED APPLICATION

This invention is based on and claims priority of Japanese patent
5 application 2001-329688, filed on October 26, 2001, the whole contents of which
are incorporated herein by reference.

BACKGROUND OF THE INVENTION

1. FIELD OF THE INVENTION

10 The present invention relates to an electronic device having a
ferroelectric layer and a method of manufacturing the same, more particularly to
an electronic device having a ferroelectric layer oriented crystallographically and
a method of manufacturing the same.

2. DESCRIPTION OF THE RELATED ART

15 A semiconductor memory, in which one memory cell is constituted of one
transistor and one capacitor, has been widely known. A capacitor of a dynamic
random access memory (DRAM) has a capacitor dielectric layer formed of a
paraelectric material. Electric charges stored in the capacitor gradually
decrease therefrom due to their leak even when the transistor is turned off.
20 Accordingly, when a voltage applied to the memory cell is removed, information
stored therein decreases and disappears before long.

A memory capable of retaining information stored therein even after
power is cut off is called a non-volatile memory. As a kind of the non-volatile
memory, a one-transistor/one-capacitor type memory, a capacitor dielectric layer
25 of which is formed of a ferroelectric material, has been known, which is called a
ferroelectric random access memory (FeRAM).

The FeRAM utilizes residual polarization of the ferroelectric material as information stored therein. The FeRAM controls a polarity of a voltage applied between a pair of electrodes of the ferroelectric capacitor, thus controlling the direction of the residual polarization. Assuming that one polarization direction be "1" and the other be "0", binary information can be stored. Since the residual polarization remains in the ferroelectric capacitor even after the applied voltage is removed therefrom, the non-volatile memory can be realized. In the non-volatile memory, information can be rewritten by a sufficient number of times, that is, 10^{10} to 10^{12} times. The non-volatile memory also has a rewriting speed of an order of several ten nanoseconds and offers a high-speed operability.

As ferroelectric materials, lead-based oxide ferroelectric materials having a perovskite structure and bismuth-based oxide ferroelectric materials having a bismuth-layered structure have been known. Typical examples of the lead-based ferroelectric materials are $\text{PbZr}_x\text{Ti}_{1-x}\text{O}_3$ (PZT), $\text{Pb}_y\text{La}_{1-y}\text{Zr}_x\text{Ti}_{1-x}\text{O}_3$ (PLZT) and the like. A typical example of the bismuth-based oxide ferroelectric materials is $\text{SrBi}_2\text{Ta}_2\text{O}_9$ (BST).

The ferroelectric capacitor offers a higher charge retention capability as the polarization of the ferroelectric material is greater, and can retain the electric potential with less capacitance. Specifically, the FeRAM can be fabricated with high integration. Furthermore, as the polarization of the ferroelectric material is greater, the polarization directions can be differentiated more clearly even at a low reading-out voltage, thus enabling the ferroelectric memory to be driven at a low voltage.

It is effective to arrange orientations of ferroelectric crystals uniformly in order to increase a polarization amount of the ferroelectric material. For example, on pages 382 to 388 of "Journal of Applied Physics" 1991, vol. 70, No.

1, disclosed is a technology of obtaining a (111)-oriented ferroelectric thin film, in which metal thin films formed of metals such as platinum (Pt) and iridium (Ir) are deposited at 500°C to obtain a (111)-oriented metal thin film, and a ferroelectric thin film such as PZT is deposited on this metal thin film at a room temperature, followed by heating of the deposited ferroelectric thin film to a range from 650°C to 700°C. However, the maximum temperature permitted for a manufacturing process of the FeRAM is usually 620°C.

The ferroelectric material such as PZT having a tetragonal simple perovskite structure has a polarization axis along the c axis $\langle 001 \rangle$. Accordingly, the polarization amount becomes maximum when the ferroelectric layer is approximately oriented along a (001) plane (hereinafter, referred to as (001)-oriented). When the ferroelectric layer is (111)-oriented, a component of the polarization produced in $\langle 001 \rangle$ direction is only about 1/1.73 in $\langle 111 \rangle$ direction that is a thickness direction of the ferroelectric layer. Although the polarization can be increased by aligning orientation, it is impossible to increase the polarization to the maximum.

SUMMARY OF THE INVENTION

An object of the present invention is to provide an electronic device capable of obtaining a ferroelectric layer having a large polarization amount and a method of manufacturing the same.

Another object of the present invention is to provide an electronic device provided with a (001)-oriented ferroelectric layer and a method of manufacturing the same.

Still another object of the present invention is to provide an electronic device provided with a ferroelectric capacitor having a ReO_3 layer as at least one

of electrodes and a method of manufacturing the same.

According to one aspect of the present invention, there is provided an electronic device including: a ReO_3 layer having a (001) orientation; and an oxide ferroelectric layer having a perovskite structure, the oxide ferroelectric layer being
5 formed on the ReO_3 layer and having a (001) orientation.

According to another aspect of the present invention, there is provided a method of manufacturing an electronic device, including the steps of: preparing a ReO_3 layer having a (001) orientation; and forming an oxide ferroelectric layer having a perovskite structure on the ReO_3 layer, the oxide ferroelectric layer
10 having a (001) orientation.

A (001)-oriented MgO layer is preferably used as an underlying layer of the ReO_3 layer.

Lattice matching can be made for the (001)-oriented ReO_3 layer and the (001)-oriented oxide ferroelectric layer having a perovskite structure; accordingly,
15 the (001)-oriented oxide ferroelectric layer having a perovskite structure can be formed on the (001)-oriented ReO_3 layer.

The MgO layer can be easily (001)-oriented. The lattice matching can be made for the (001)-oriented MgO layer and the (001)-oriented ReO_3 layer. Hence, the (001)-oriented ReO_3 layer and the (001)-oriented oxide ferroelectric
20 layer having a perovskite structure can be formed on the (001)-oriented MgO layer sequentially.

The term " ReO_3 " used herein includes ReO_3 to which metal other than Re is added, for example, for controlling a lattice constant thereof.

In such a manner as described above, it is possible to form a ferroelectric
25 capacitor capable of realizing greater polarization.

BRIEF DESCRIPTION OF THE DRAWINGS

Fig. 1A is a schematic cross-sectional view of an electronic device; Fig. 1B is a schematic block diagram showing a constitution of a metalorganic chemical vapor deposition (MOCVD) apparatus; Fig. 1C is a schematic cross-sectional view showing an upper electrode of the electronic device when a stacked structure is adopted therefor; and Fig. 1D is a schematic cross-sectional view of the electronic device when a single crystal MgO layer is used therefor, all of which are made for illustrating embodiments of the present invention.

Figs. 2A and 2B are structural views showing chemical formulae of $\text{Mg}(\text{DPM})_2$ and i-PrO .

Figs. 3A and 3B are cross-sectional views of constitutional examples of an electronic device having a ferroelectric capacitor according to embodiments of the present invention.

DESCRIPTION OF THE PREFERRED EMBODIMENTS

Hereinafter, description will be made on embodiments of the present invention with reference to the drawings.

Fig. 1A shows a structure of a ferroelectric capacitor according to a fundamental embodiment of the present invention. A silicon oxide layer 11 is formed on a Si substrate 10. The silicon oxide layer 11 can be formed by thermal oxidation of silicon, chemical vapor deposition (CVD) or the like. The silicon oxide layer 11 may be formed by other methods. The silicon oxide layer 11 has an amorphous phase. A (001)-oriented MgO layer 12 is formed on the silicon oxide layer 11, a (001)-oriented ReO_3 layer 13 is formed on the MgO layer 12, and a (001)-oriented PZT layer 14 is formed on the ReO_3 layer 13.

The (001)-oriented MgO layer 12, the (001)-oriented ReO_3 layer 13 on

the MgO layer 12 and the (001)-oriented PZT layer 14 on the ReO₃ layer 13, the PZT layer 14 being a ferroelectric layer having a perovskite structure, can be deposited by metalorganic chemical vapor deposition (MOCVD) using a metalorganic (MO) material.

5 Fig. 1B schematically shows a structure of an apparatus for depositing a film by MOCVD. A liquid container 21-1 contains a metalorganic material solution used for deposition. Pressurized He gas is fed to the liquid container 21-1 from a pipe opened to a space on the solution, thus enabling the solution to be supplied to another pipe 22-1 deeply intruded into the solution. A flow rate of
10 the supplied solution is controlled by a mass flow controller (MFC) 24-1, and the solution is supplied to a vaporizer 27-1 through a pipe 25-1.

A carrier gas pipe 26 is connected to the vaporizer 27-1. The liquid raw material solution supplied to the vaporizer 27-1 together with carrier gas N₂ is vaporized by the vaporizer 27-1 and supplied to a pipe 28-1.

15 A liquid container 21-2, a pipe 22-2, a mass flow controller 24-2, a pipe 25-2, a vaporizer 27-2 and a pipe 28-2 have similar structures as those of the liquid container 21-1, the pipe 22-1, the mass flow controller 24-1, the pipe 25-1, the vaporizer 27-1 and the pipe 28-1, which are described above, respectively. Furthermore, any number of similar raw material supply systems may be
20 provided.

The vaporizer 27-1 may be connected with other liquid raw material supply systems having similar structures as those of the liquid container 21-1, the pipe 22-1, the mass flow controller 24-1 and the pipe 25-1. Other vaporizers can also be provided with any number of the liquid raw material supply systems.

25 A reaction chamber 30 has raw material pipes such as a gas pipe 29 and the liquid raw material pipes 28-1, 28-2..., and can supply raw material gas from

a showerhead 32. A susceptor 34 capable of controlling a temperature thereof is disposed at a lower portion of the reaction chamber 30, and a substrate 35 composed of, for example, a silicon substrate provided with a silicon oxide layer is disposed on the susceptor 34.

5 In the above description, an example is shown, in which the raw material supply systems are provided in plural; however, a single system may be employed. Moreover, an example of a single reaction chamber is shown; however, a plurality of the reaction chambers may be provided.

With regard to a metalorganic material contained in the liquid containers,
10 for example, as a Mg raw material, a solution obtained by dissolving $\text{Mg}(\text{DPM})_2$ (where DPM is dipivaloilmethanate) in the tetrahydrofuran (THF) can be used.

Fig. 2A is a chemical formula showing a chemical structure of $\text{Mg}(\text{DPM})_2$. Dipivaloilmethanate (DPM) is bonded at each side of a Mg atom. DPM is monovalent, and n pieces of DPMs can be bonded to an n -valent atom.

15 As a Re material, a solution obtained by dissolving $\text{Re}(\text{DPM})_2$ in THF can be used. A chemical formula of $\text{Re}(\text{DPM})_2$ is equivalent to that obtained by replacing Mg with Re in the chemical formula shown in Fig. 2A.

As a Pb material, a solution obtained by dissolving $\text{Pb}(\text{DPM})_2$ in THF can be used. A structure of $\text{Pb}(\text{DPM})_2$ is equivalent to that obtained by replacing Mg
20 with Pb in the structure shown in Fig. 2A.

As a Zr material, a solution obtained by dissolving $\text{Zr}(\text{DPM})_4$ in THF can be used. $\text{Zr}(\text{DPM})_4$ has a structure where four DPMs are bonded around one Zr atom.

As a Ti material, a solution obtained by dissolving $\text{Ti}(\text{i-PrO})_2(\text{DPM})_2$
25 (where i-PrO is an iso-proxy group) in THF can be used. A structure of $\text{Ti}(\text{i-PrO})_2(\text{DPM})_2$ is equivalent to that obtained by replacing Mg with Ti in the

structure shown in Fig. 2A and by bonding two iso-proxy groups shown in Fig. 2B to Ti. Note that the metalorganic (MO) material is not limited to these examples.

In order to deposit the MgO layer 12 shown in Fig. 1A, pressurized
5 helium (He) gas is fed to the liquid containers 21 containing the solution obtained by dissolving Mg(DPM)_2 in THF, and the solution is made to pass through the vaporizers 27 heated at 260°C , vaporized, and loaded on the carrier gas N_2 .

The Mg raw material, for which N_2 is used as carrier gas, is fed through the pipes 28 to the showerhead 32, and supplied to the silicon oxide film on the
10 substrate 35 together with O_2 gas supplied from the pipe 29. The silicon oxide film is heated to 560°C , decomposes the supplied metalorganic gas, and combines the decomposed gas with oxygen, thus depositing a (001)-oriented MgO layer. A thickness of the (001)-oriented MgO layer is set, for example, in a range from 50 to 100 nm.

15 Deposition temperature is not limited to 560°C . Preferably, deposition is carried out with substrate temperature of 620°C or lower. Accordingly, a step of the deposition can be harmonized with other manufacturing steps for the FeRAM device.

Next, description will be made for the case of depositing the ReO_3 layer
20 13 on the (001)-oriented MgO layer 12. In order to deposit the ReO_3 layer 13, the liquid raw material obtained by dissolving Re(DPM)_2 in THF, which is contained in the liquid containers 21, is used, and the metalorganic material loaded on the carrier gas is fed to the showerhead 32 in the same manner as the above-described process. To the showerhead 32, O_2 gas, mixed gas of O_2 gas
25 and N_2 gas or the like is simultaneously supplied.

The substrate 35 having the (001)-oriented MgO layer 12 formed thereon

is kept at a constant temperature of 560°C by means of the susceptor 34. The raw material gas is supplied onto the (001)-oriented MgO layer 12 kept at 560°C, whereby the (001)-oriented ReO_3 layer 13 is deposited. A thickness of the (001)-oriented ReO_3 layer 13 is set, for example, in a range from 20 to 50 nm.

5 After the (001)-oriented ReO_3 layer 13 is deposited, the PZT layer 14 is deposited thereon. For the PZT, as a Pb raw material, the solution obtained by dissolving $\text{Pb}(\text{DPM})_2$ in THF is used; as a Zr raw material, the solution obtained by dissolving $\text{Zr}(\text{DPM})_4$ in THF is used; and as a Ti raw material, the solution obtained by dissolving $\text{Ti}(\text{i-PrO})_2(\text{DPM})_2$ in THF is used. Pressurized helium
10 gas is fed to three liquid containers containing these liquid raw materials, and the liquid raw materials are vaporized by one or three vaporizers and supplied to the showerhead 32.

The substrate temperature is kept at 560°C, and $\text{Pb}(\text{DPM})_2$ gas, $\text{Zr}(\text{DPM})_4$ gas, $\text{Ti}(\text{i-PrO})_2(\text{DPM})_2$ gas and oxygen are simultaneously blown onto
15 the substrate, thus the $\text{Pb}(\text{Zr, Ti})\text{O}_3$ (PZT) layer 14 is deposited on the (001)-oriented ReO_3 layer 13. The deposited PZT layer 14 has also (001) orientation. A thickness of the (001)-oriented PZT layer 14 is set, for example, in a range from 80 to 150 nm.

As described above, an MgO layer is deposited on an amorphous silicon
20 oxide layer 11 by MOCVD, to obtain a (001)-oriented MgO layer 12. On the (001)-oriented MgO layer 12, there can be deposited a ReO_3 layer 13, which is (001)-oriented in accordance with the orientation of the underlying layer, that is, the MgO layer 12. Furthermore, on the (001)-oriented ReO_3 layer 13, there can be deposited the PZT layer 14, which is (001)-oriented in accordance with the
25 orientation of the underlying layers, that is, the MgO layer 12 and the ReO_3 layer 13.

An upper electrode 15 is formed on the PZT layer 14. The upper electrode 15 is not required to be (001)-oriented and can be formed of an electrode material publicly known hitherto. For example, an IrO_2 layer is deposited by MOCVD. In this case, as an Ir raw material, a solution obtained by
5 dissolving $\text{Ir}(\text{DPM})_3$ in THF is used. Process for vaporizing the material is similar as that described above. The substrate temperature is kept at 560°C , and $\text{Ir}(\text{DPM})_3$ gas and oxygen are simultaneously blown thereonto, thus enabling the upper electrode 15 made of IrO_2 , which is also referred to as an IrO_2 layer, to be deposited on the PZT layer 14. A thickness of the IrO_2 layer 15 is set, for
10 example, in a range from 100 to 150 nm.

Description has been made for the case of forming the upper electrode 15 of an IrO_2 layer; however, various materials can be used for the upper electrode irrespective of the orientation of the ferroelectric layer.

As shown in Fig. 1C, for the upper electrode, a stacked layer 15 obtained
15 by stacking an IrO_2 layer 15-1 and a SrRuO_3 layer 15-2 may be used. Deposition methods other than MOCVD may also be used.

For example, the IrO_2 layer 15-1 can be deposited by sputtering using an IrO_2 target. In this case, the substrate is kept at a room temperature, and the target is sputtered by use of work gas Ar at a vacuum degree of 3×10^{-4} Torr, thus
20 the IrO_2 layer 15-1 is deposited. A thickness of the IrO_2 layer 15-1 is set, for example, in a range from 100 to 150 nm.

The SrRuO_3 layer 15-2 to be deposited on the IrO_2 layer 15-1 can also be deposited by sputtering. SrRuO_3 is used as a target, the substrate is kept at a room temperature, the vacuum degree is set at 3×10^{-4} Torr, and Ar is used as
25 work gas. Under the above-described conditions, the target is sputtered, and thus the SrRuO_3 layer 15-2 is deposited. A thickness of the SrRuO_3 layer 15-2

is set, for example, in a range from 10 to 30 nm.

Description has been made above for the case of using PZT as a ferroelectric material; however, other oxide ferroelectric materials having a perovskite structure can be employed. For example, $\text{Pb}_y\text{La}_{1-y}\text{Zr}_x\text{Ti}_{1-x}\text{O}_3$ (PLZT),
5 $\text{Pb}_{1-a-b-c}\text{La}_a\text{Sr}_b\text{Ca}_c\text{Zr}_{1-x}\text{Ti}_x\text{O}_3$ (PLSCZT) and the like can be used.

Moreover, description has been made for the case of using only O_2 gas as a kind of gas. However, mixed gas of O_2 and other gas, for example, O_2/N_2 , O_2/Ar , O_2/He and $\text{O}_2/\text{N}_2\text{O}$, can also be used.

ReO_3 added with a small amount of other metal shows an electrical
10 resistivity of an order of $10^{-6} \Omega\cdot\text{m}$ at 300°K . A metal layer used as an electrode can be utilized effectively as long as an electrical resistivity thereof is $10^{-5} \Omega\cdot\text{m}$ or less. Accordingly, ReO_3 added with the other metal (metal impurities) can be utilized effectively as such an electrode of the ferroelectric capacitor.

Note that the MgO layer is deposited on the amorphous silicon oxide
15 layer 11, thus forming the (001)-oriented MgO layer 12; however, it will be obvious that a (001) plane of single crystal MgO can be used in place of the deposited MgO layer.

Fig. 1D shows the case where a ReO_3 layer 13 and a ferroelectric layer
14 having a perovskite structure are epitaxially grown in this order on a single
20 crystal MgO layer 12 having a (001) plane, and then an upper electrode 15 is formed on the ferroelectric layer 14.

Furthermore, it will be possible to deposit the (001)-oriented MgO layer
12, ReO_3 layer 13 and ferroelectric layer 14 by, in place of CVD using the metalorganic (MO) raw materials, CVD using other raw materials. Similarly, it
25 will be possible to deposit the above (001)-oriented layers by sputtering.

The ferroelectric layer 14 is (001)-oriented, thus enabling the polarization

caused by application of the voltage to be aligned to a direction perpendicular to the electrode surface. Therefore, it is made possible to utilize the polarization of the ferroelectric layer most effectively.

Figs. 3A and 3B show constitutional examples of electronic devices, each
5 using the ferroelectric capacitor as described above.

Fig. 3A shows an example where electrodes are taken out of upper and lower surfaces of a ferroelectric capacitor. An element isolation region 40 is formed on a surface of a Si substrate 10 by shallow trench isolation (STI). Two MOS transistors are formed in an active region defined by the element isolation
10 region 40. The two MOS transistors have one source/drain region 46 as a common region and other source/drain regions 45 on both sides thereof, which are connected with the ferroelectric capacitors, respectively.

On a channel region between the source/drain regions, is disposed an insulated gate electrode formed of a gate insulating film 41, a polycrystalline gate
15 electrode 42 and a silicide gate electrode 43. A side spacer 44 is formed on a sidewall of the insulated gate electrode. An amorphous insulating layer 11 made of silicon oxide or the like is formed over surfaces where the semiconductor devices are formed. Furthermore, a (001)-oriented MgO layer 12 is formed on a surface of the amorphous insulating layer 11.

20 In order to form an extraction electrode for each of the both-side source/drain regions 45, a contact hole is formed through the MgO layer 12 and the amorphous insulating layer 11. An extraction plug composed of, for example, barrier metal 48 and a tungsten (W) plug 49 is formed in the contact hole. Then, unnecessary electrode layers on the MgO layer 12 are removed by, for example,
25 chemical mechanical polishing (CMP). Subsequently, on the MgO layer 12, is formed a ferroelectric capacitor composed of the lower ReO_3 layer 13, the

ferroelectric layer 14 having a perovskite structure 14 and the upper electrode 15.

The MgO layer 12 is (001)-oriented, thus making it possible to form the (001)-oriented lower ReO_3 layer 13 and the (001)-oriented ferroelectric layer 14 having a perovskite structure.

5 After forming the ferroelectric capacitor, an insulating layer 50 made of silicon oxide or the like is deposited to cover a surface thereof. Moreover, a contact hole is formed through the insulating layer 50, and then a barrier metal layer 51 and a metal conductive layer 52 made of W or the like are buried in the contact hole, thus the extraction electrode is formed. After forming the
10 extraction electrode, unnecessary electrode layers on the insulating layer 50 are removed, and upper wirings 54 and 55 are formed. Surfaces of the upper wirings 54 and 55 are covered with an insulating layer 60.

Fig. 3B shows a constitution, in which two electrodes are taken out of the upper surface of the ferroelectric capacitor. An element isolation region 40 of
15 silicon oxide formed by local oxidation of silicon (LOCOS) is formed on the surface of the Si substrate 10. One MOS transistor is formed in an active region defined by the element isolation region 40.

On a channel region, is disposed an insulated gate electrode formed of a gate insulating film 41, a polycrystalline gate electrode 42 and a polycrystalline
20 silicide gate electrode 43. A side spacer 44 is formed on a sidewall of the insulated gate electrode. Source/drain regions 45 and 46 are formed on both sides of the gate electrode by ion implantation and the like.

An amorphous insulating layer 48 made of silicon oxide or the like is formed to cover the MOS transistor. Plugs 49 for deriving the source/drain
25 regions 45 and 46 are formed. A silicon nitride layer 59, for example, having an amorphous phase is formed on a surface of the amorphous insulating layer 48

through which the plugs 49 are formed, thus an oxygen shielding layer is formed.

On the amorphous silicon nitride layer 59, a (001)-oriented MgO layer 12 is formed. It is conceivable that the (001)-oriented MgO layer 12 can be deposited as long as its underlying layer is amorphous. On the (001)-oriented
5 MgO layer 12, is formed a ferroelectric capacitor composed of a (001)-oriented ReO₃ layer 13, a (001)-oriented ferroelectric layer 14 having a perovskite structure and an upper electrode 15. The lower ReO₃ electrode 13 is extracted along a direction perpendicular to the drawing sheet. An insulating layer 18 made of silicon oxide or the like is formed to cover the ferroelectric capacitor.

10 Desired portions of the insulating layer 18, MgO layer 12 and silicon nitride layer 59 are removed by etching, to form contact holes. Then, a local wiring 19 connects the plug 49 exposed in the contact hole with the upper electrode 15. An insulating layer 50 is further formed to cover the local wiring 19. Through the insulating layer 50, an opening for exposing the plug 49 on the other
15 source/drain region 46 is formed. The other wiring 55 is formed ,filling the opening.

The above-described constitutions around the ferroelectric capacitor and around the transistor, which are shown in Figs. 3A and 3B, respectively, are examples, and have no limitative meaning. Various alternations and exchanges
20 may be employed. Multi-layered wiring structure can be formed by other publicly known techniques. As described above, the electronic device with the ferroelectric capacitor, for example, a semiconductor integrated circuit device can be manufactured.

Although the present invention has been described along the
25 embodiments, the present invention is not limited thereto. It will be obvious to those skilled in the art that various modifications, improvements and

combinations can be made.